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# Thermal diffusivity and thermal conductivity of dispersed glass sphere composites over a range of volume fractions

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## Abstract

Glass spheres are often used as filler materials for composites. Comparatively few articles in the literature have been devoted to the measurement or modelling of thermal properties of composites containing glass spheres, and there does not appear to be any reported data on the measurement of thermal diffusivities over a range of filler volume fractions. In this study, the thermal diffusivities of guar-gel/glass sphere composites were measured using a transient comparative method. The addition of the glass beads to the gel increased the thermal diffusivity of the composite, more than doubling the thermal diffusivity of the composite relative to the diffusivity of the gel at the maximum glass volume fraction of approximately 0.57. Thermal conductivities of the composites were derived from the thermal diffusivity measurements, measured densities and estimated specific heat capacities of the composites. Two approaches to modelling the effective thermal diffusivity were considered.

**Keywords:** thermal diffusivity, thermal conductivity, glass spheres, composites

## 1. Introduction

Composite materials offer advantages over single-component, homogeneous materials in that desirable properties of two or more component materials may be combined to produce a material that may be customised for a given application. For example, metal powders may be added to

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polymeric materials to enhance the thermal and mechanical properties of the material, while retaining the mouldable, extrude-able nature of polymers [1-16]. Hollow or solid glass spheres have also been used as a filler material in resinous or polymeric materials [17-23]. Glass spheres can improve mechanical properties, without adding significantly to the weight of the composites, and do not enhance thermal conductivity to the same extent as metal fillers, which may sometimes be desirable [18-20]. However, comparatively less attention has been devoted to the measurement and modelling of thermal properties of this class of material than for metal-filled composites.

Zwart and Yovanovich [24] modelled the thermal diffusivity of simply packed glass spheres, but not as part of a composite material. Liang and Li [18] measured the thermal conductivity of hollow glass-bead-filled polypropylene composites for use in the building sector. They found that the composites provided adequate mechanical properties while improving their sound and thermal insulating qualities. Mishra et al. [20] measured the thermal conductivity of solid glass spheres dispersed in polymer matrices, for volume fractions of glass up to 0.27, and observed that the addition of the glass reduced the thermal conductivity of the composite relative to the conductivity of the polymer. There does not appear to be any data for thermal diffusivity of composites containing dispersions of glass spheres; and certainly none covering a wide range of filler volume fractions. Such data is useful for analysing the effects of filler volume on thermal properties, particularly for the purposes of validating effective thermal property models. The aim of this work was to measure the effective thermal diffusivity of mono-disperse glass spheres within a gel matrix over the full range of filler volume (i.e. from zero to the packing factor limit).

## **2. Materials and Methods**

### *2.1 Materials*

1 In order to measure thermal diffusivity of glass-sphere dispersions for a range of filler volumes, a  
2 suitable matrix material was required. Guar gel was chosen as it sets at room temperature and has a  
3 relatively slow setting time that allows for the glass spheres (which are more dense than water) to  
4 be suspended with approximately uniform spatial distribution. The guar gels were made of 4 %  
5 (mass basis) guar gel powder, and 96 % water, with trace amounts of borax ( $\text{Na}_2\text{B}_4\text{O}_7 \cdot 10\text{H}_2\text{O}$ ) to  
6 serve as a gelling agent. The glass spheres had approximately 3.5 mm diameters and the test  
7 samples were contained in spherical aluminium containers with radii of approximately 75 mm.

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9 The maximum volume fraction of glass that was possible was approximately 0.57. Unlike samples  
10 having lower glass fractions, these samples were prepared by first filling the sample container with  
11 the glass beads and subsequently pouring the unset gel solution into the container allowing it to fill  
12 the interstitial regions between the glass sphere and the gaps between the spheres and the walls.  
13 This packing fraction of 0.57 is much smaller than the theoretical maximum packing factor for  
14 monodisperse spheres (0.74) [32]. Nielsen [32] quoted fractions of 0.6 to 0.63 for random packings  
15 of spheres; however, the fact that the sample containers themselves were spherical would have  
16 meant that the effect of the curved wall on the packing factor would have been more influential  
17 than if it had been a plane walled container.

## 42 *2.2 Methods*

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44 The thermal diffusivity was measured using a transient comparative method that has been described  
45 previously [13]. Briefly, the method involves heating or cooling two samples having the same  
46 geometry (spheres in this case), one of which is a control or reference sample having known thermal  
47 properties while the other is the test sample of unknown thermal properties. The logarithms of the  
48 dimensionless temperature changes ( $\theta$ ) defined by Eq. (1) of the samples are plotted against time  
49 (Figure 1).  
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$$\theta = \frac{T(t) - T_{\infty}}{T_i - T_{\infty}} \quad (1)$$

The linear portion of the temperature histories may be fitted by Eq. (2):

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$$\ln \theta = F - st \quad (2)$$

where  $s$  is the slope and  $F$  is the intercept. By comparison with the analytical solution for transient conduction in a sphere for a Fourier number greater than 0.2 (Chapter4, [25]), it can be seen that  $s$  is related to the sample's thermal diffusivity according to Eq. (3):

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$$s = \frac{\lambda^2 \alpha}{R^2} \quad (3)$$

where  $\alpha$  is thermal diffusivity,  $R$  is the radius of the sample and  $\lambda$  is the first root of Eq. (4):

$$\lambda \cot \lambda + Bi - 1 = 0 \quad (4)$$

where  $Bi$  is the Biot number:

$$Bi = \frac{hR}{k} \quad (5)$$

where  $h$  is the heat transfer coefficient around the sample, and  $k$  is the thermal conductivity of the sample. Provided the radii of the test and control samples are the same and Biot numbers of the samples are high enough (e.g. > 100) the ratio of the slopes of the linear portions of the temperature history is directly proportional to the ratios of the thermal diffusivity [13], i.e:

$$\frac{\alpha_e}{\alpha_c} = \frac{s_e}{s_c} \quad (6)$$

where the subscript  $e$  refers to the test sample and the subscript  $c$  refers to the control or reference sample.

Heat transfer coefficients were estimated using back-calculation from cooling histories of a copper sphere using the lumped heat capacity approximation [25].

1 In this study, a sample of guar gel that did not contain any glass spheres was used as the reference,  
2 while the test samples contained varying quantities of glass spheres dispersed uniformly throughout  
3 the sample volume. The samples were initially equilibrated at approximately 20 °C before being  
4 cooled in an ice/water bath. A schematic of the measurement apparatus is shown in Figure 2. All  
5 measurements were replicated at least once.  
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### 11 **3. Results and Discussion**

12 Figure 3 shows a plot of the mean thermal diffusivities of the test samples relative to the thermal  
13 diffusivity of the guar gel reference material ( $1.4 \times 10^{-7} \text{ m}^2 \text{ s}^{-1}$ ), i.e.  $\alpha_e/\alpha_c$ . There is a clear increase in  
14 thermal diffusivity with increasing filler volume, to the extent that the composite with the maximum  
15 filler volume has a thermal diffusivity greater than twice that of the control sample. The error bars in  
16 Fig. 3 are based on the standard deviation between replicate measurements at each volume fraction  
17 (on average 1.5 % of the mean value at each volume fraction).  
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33 As mentioned in the Methods section, the validity of Eq. (6) relies on the Biot number being high  
34 enough such that the  $\lambda$  roots are asymptotic. In order to test this assumption it was necessary to  
35 calculate the thermal conductivity of the samples so that the Biot number may be evaluated by Eq.  
36 (5). The thermal conductivity was determined from a rearrangement of the definition of thermal  
37 diffusivity [25], i.e.:

$$38 k = \alpha \rho c_p \quad (7)$$

39 where  $\rho$  is the density of the sample and  $c_p$  is its heat capacity. The density of the glass spheres was  
40 measured via Archimedes' principle to be  $2520 \text{ kg m}^{-3}$ , and the density of the guar gel was  $1015 \text{ kg}$   
41  $\text{m}^{-3}$ . Since the volumes of the spherical sample containers were known, measurement of the test  
42 samples' densities were determined simply by weighing. The density of the test samples increased  
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1 linearly with the increase in the volume of glass added to the gel (Figure 4). The measured densities  
2 where compared to the densities predicted by the Eq. (8):  
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$$4 \rho_e = v_{gel}\rho_{gel} + v_{glass}\rho_{glass} \quad (8)$$

7 The difference between measured densities and densities predicted by Eq. (8) was attributed to the  
8 presence of air. The average value of the estimated volume fraction of air was 0.03. This was  
9 considered small enough to be negligible.  
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11 Direct measurement of the test samples' specific heat capacity was not practical due to the large  
12 sizes of the samples. Instead, they were estimated via Eq. (9):  
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$$14 c_{P,e} = x_{gel}c_{P,gel} + x_{glass}c_{P,glass} \quad (9)$$

15 The specific heat capacity of the guar gel as measured by differential scanning calorimetry was 4150  
16 J kg<sup>-1</sup> K<sup>-1</sup>. The mass fractions ( $x$ ) of the glass and the gels could be determined from the volume  
17 fractions and densities of the samples using Eq. (10):  
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$$19 x_{glass} = \frac{\rho_{glass}v_{glass}}{\rho_e} \quad (10)$$

20 Since the supplier of the glass beads did not provide specific heat capacity data, a mid-range value of  
21 800 J kg<sup>-1</sup> K<sup>-1</sup> for glass from the literature was used [25-27]. Figure 5 shows the specific heat capacity  
22 of the test samples estimated using Eq. (9) based on the estimated specific heat value of 800 J kg<sup>-1</sup> K<sup>-1</sup>  
23 <sup>1</sup>, as well dashed lines indicating the  $\pm 10\%$  uncertainty range of the estimates. It is clear that a 10 %  
24 uncertainty in the specific heat capacity of the glass contributes less than 10 % uncertainty to  $c_{P,e}$ .  
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26 The thermal conductivities of the test samples were calculated using Eq. (7) from the measured  
27 thermal diffusivity data, the measured density data and the specific heat capacities estimated by Eq.  
28 (8). The thermal conductivity was then used to calculate the Biot number, which in turn could be  
29 used to calculate  $\lambda$  using Eq. (4). The heat transfer coefficient measurements ranged between 1200  
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and  $1500 \text{ W m}^{-2} \text{ K}^{-1}$ , which produced  $Bi > 100$  for all the samples involved, thereby justifying the use of Eq. (6).

Figure 6 shows the thermal conductivity of the test samples as a function of the volume fraction of glass relative to the thermal conductivity of the guar gel ( $0.57 \text{ W m}^{-2} \text{ K}^{-1}$ ). The error bars in Figure 5 are based on the estimated uncertainty in the thermal conductivity data of  $\pm 5\%$ , which is based on combining the error estimates from the thermal diffusivity, specific heat capacity and density measurements. The increase in thermal conductivity with the addition of glass spheres is not as significant as for thermal diffusivity. This is due to the decrease in specific heat capacity with increasing glass volume that offsets both the increase in thermal conductivity with increasing glass volume and the increase in density with increasing glass volume.

Figure 6 also shows predictions of the Series (Eq. 11) and Parallel (Eq. 12) effective thermal conductivity models which form the upper and lower limits to the thermal conductivity of a heterogeneous material [28], based on an estimated thermal conductivity of the glass spheres of  $1.1 \text{ W m}^{-2} \text{ K}^{-1}$  [25-27].

$$k_{Series} = \frac{1}{\frac{v_{glass}}{k_{glass}} + \frac{v_{gel}}{k_{gel}}} \quad (11)$$

$$k_{Parallel} = k_{glass}v_{glass} + k_{gel}v_{gel} \quad (12)$$

Two other models are also plotted in Figure 6, the Maxwell-Eucken model with glass as the dispersed phase [28]:

$$k_e = k_{gel} \frac{2k_{gel} + k_{glass} - 2(k_{gel} - k_{glass})v_{glass}}{2k_{gel} + k_{glass} + (k_{gel} - k_{glass})v_{glass}} \quad (13)$$

and the Lewis-Nielsen model [32-34]:

$$k_e = k_{gel} \frac{1 + ABv_{glass}}{1 - B\psi v_{glass}} \quad (14)$$

$$A = k_E - 1 \quad (15)$$

$$B = \frac{\frac{k_{glass}}{k_{gel}} - 1}{\frac{k_{glass}}{k_{gel}} + A} \quad (16)$$

$$\psi = 1 + \left[ \frac{1 - v_{max}}{v_{max}^2} \right] v_{glass} \quad (17)$$

where  $k_E$  is 2.5 for rigid spheres [32,33] and  $v_{max}$  for these experiments (the maximum filler volume fraction was 0.57). At lower volume fractions of the glass, the thermal conductivity is modelled well by the Series model, while the Maxwell-Eucken and Lewis Nielsen models also lie within the measurement error bars. As the filler volume increases the measured data move away from the Series model and the Maxwell-Eucken model fits the data more closely.

Surprisingly, there do not appear to be many simple effective thermal diffusivity models in the literature where effective thermal diffusivity may be predicted from components' thermal diffusivities and compositions (i.e. thermal diffusivity equivalents of Eqs. 11 and 12). Instead effective thermal diffusivities tend to be modelled based on effective thermal conductivity, effective specific heat capacity and effective density models (e.g. [29,30]), i.e.:

$$\alpha_e = \frac{k_e}{\rho_e c_{P,e}} \quad (18)$$

However, Eq. (18) only provides approximations of the effective thermal conductivity of composite materials [35,36], and potentially it would be more accurate to model effective diffusivity directly if the model predictions may be compared to directly measured thermal diffusivity data, rather than data derived from thermal conductivity, specific heat capacity and density using Eq. (18). For example, in this study the use of Eq. (18) to model the measured thermal diffusivity data incorporates the uncertainty in the data for thermal conductivity and specific heat capacity of the glass (which were not measured, but taken from the literature), as well as the uncertainty involved in the effective thermal conductivity and specific heat capacity models. Also, the ratio of the thermal conductivity of the glass to the thermal conductivity of the gel is 1.9; whereas the ratio of the thermal diffusivity of the glass to the thermal diffusivity of the gel is 8.9; hence there is inherently



1 more uncertainty in predicting the effective thermal diffusivity than the effective thermal  
2 conductivity [31].  
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5 Using the approach that has been applied to thermal conductivity, a possible effective thermal  
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7 diffusivity model could be the weighted mean of the components' thermal diffusivities, as has been  
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9 suggested previously [37]. Since the harmonic mean (i.e. Series model, Eq.11) fitted the thermal  
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11 conductivity better than the arithmetic mean, a harmonic mean was used to model thermal  
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13 diffusivity of the composites (Eq. 19):  
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$$\alpha_e = \frac{1}{\frac{v_{glass}}{\alpha_{glass}} + \frac{v_{gel}}{\alpha_{gel}}} \quad (19)$$

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22 Equation 19 is plotted in Figure 3, where it can be seen that it fits the measured thermal diffusivity  
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24 to some extent, particularly for lower volume fractions of glass. However, unlike the case of thermal  
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26 conductivity where Eqs. (11) and (12) may be derived from analysis of steady-state conduction  
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28 resistances in series and parallel, Eq. (19) has no theoretical basis, and is merely an average.  
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32 Figure 3 also shows the predictions from Eq. (20), which incorporates the Series thermal conductivity  
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34 model (Eq. 11) into Eq. (18) and  $\rho_e$  and  $c_{p,e}$  were determined from Eqs. (8) and (9) respectively:  
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$$\alpha_e = \frac{k_{Series}}{\rho_e c_{p,e}} \quad (20)$$

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41 Equation (20) also provides reasonable fits to the measured data; however, it is worth observing  
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43 from Figure 3 that Eqs. (19) and (20) produce different results because they are not mathematically  
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45 equivalent. It is clear, then, that effective thermal conductivity models cannot be converted directly  
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47 to effective thermal diffusivity models simply by substituting  $\alpha$  for  $k$ . The modelling task is  
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49 complicated by the fact that thermal conductivity is defined from a steady-state heat transfer  
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51 analysis, while thermal diffusivity is defined from a transient heat transfer analysis. During transient  
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53 heat transfer the effective thermal conductivity of a heterogeneous or composite material may vary  
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55 significantly with time as a temperature 'front' progresses between the centre and the surface of an  
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object. Therefore, to be more accurate Eq. (18), should incorporate a time-averaged value for  $k_e$ , rather than a steady-state value. It appears there is scope more work to be done in this area.

#### 4. Conclusion

The effective thermal diffusivities of dispersions of glass spheres within a guar gel matrix were measured over a range of volume fractions. The addition of the glass beads to the gel increased the thermal diffusivity of the composite, more than doubling the thermal diffusivity of the composite relative to the diffusivity of the gel at the maximum glass volume fraction of approximately 0.57. The thermal conductivity of the composite was derived from the thermal diffusivity measurements, measured densities and estimated specific heat capacities of the composites. Of the two approaches that were considered for modelling the effective thermal diffusivity, neither is entirely satisfactory and there is scope for more work on this problem.

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## Nomenclature

A parameter defined by Eq. (14)

1	$B$	parameter defined by Eq. (15)
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3	$Bi$	Biot number defined by Eq. (5)
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6	$c_p$	specific heat capacity ( $\text{J kg}^{-1} \text{K}^{-1}$ )
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9	$F$	intercept of linear portion of temperature history
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12	$h$	heat transfer coefficient ( $\text{W m}^{-2} \text{K}^{-1}$ )
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15	$k$	thermal conductivity ( $\text{W m}^{-1} \text{K}^{-1}$ )
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18	$R$	radius (m)
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21	$s$	slope of linear portion of temperature history ( $\text{s}^{-1}$ )
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24	$t$	time (s)
25		
26		
27	$T$	temperature ( $^{\circ}\text{C}$ )
28		
29		
30	$v$	volume fraction
31		
32		
33		
34	$x$	mass fraction
35		
36		
37		
38		
39		
40	$\alpha$	thermal diffusivity ( $\text{m}^2 \text{s}^{-1}$ )
41		
42		
43	$\theta$	dimensionless temperature change
44		
45		
46	$\lambda$	roots of Eq. (4)
47		
48		
49	$\psi$	parameter defined by Eq. (14)
50		
51		
52		
53		
54		
55	Subscripts	
56		
57		
58	$\infty$	bulk condition
59		
60		
61		
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1	<i>c</i>	referring to reference sample
2		
3	<i>e</i>	referring to test sample or effective property
4		
5		
6	gel	referring to the guar gel
7		
8		
9	glass	referring to the glass spheres
10		
11		
12	<i>i</i>	initial value
13		
14		
15	<i>max</i>	maximum value
16		
17		
18		
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20		

## Figure Captions

Figure 1: Temperature histories for cooling spheres

Figure 2: Experimental apparatus

Figure 3: Measured diffusivity ratios with effective diffusivity model predictions

Figure 4: Measured density ratios for varying glass filler fractions

Figure 5: Estimated specific heat capacity of gel/glass composite for varying glass filler fractions.

Figure 6: Thermal conductivity data derived from thermal diffusivity measurements with effective thermal conductivity model predictions.

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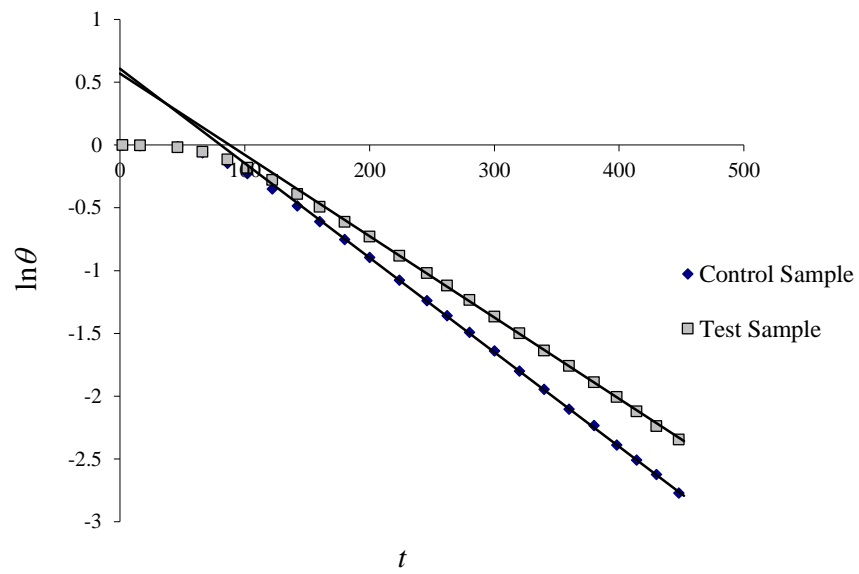


Figure 1

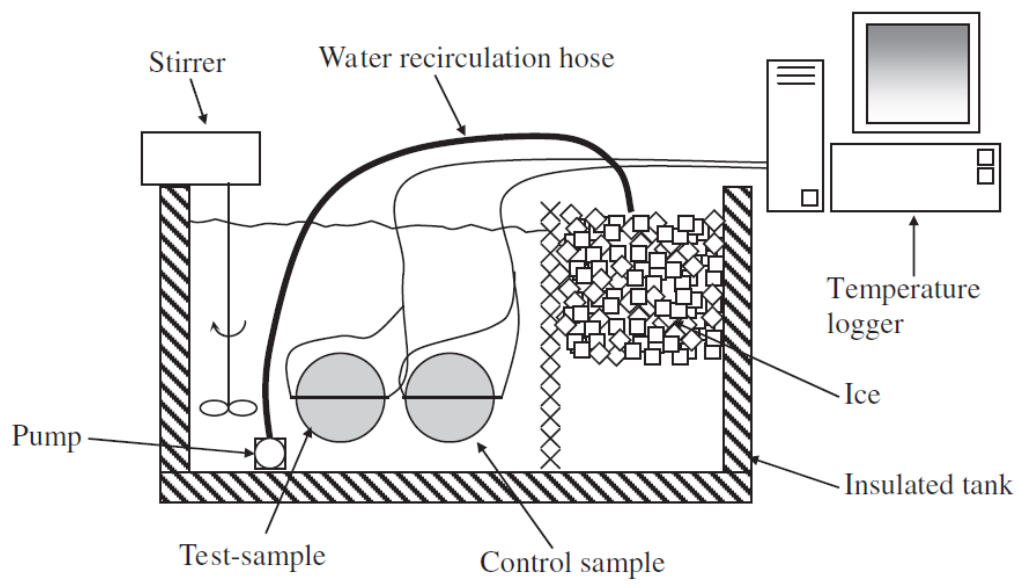


Figure 2



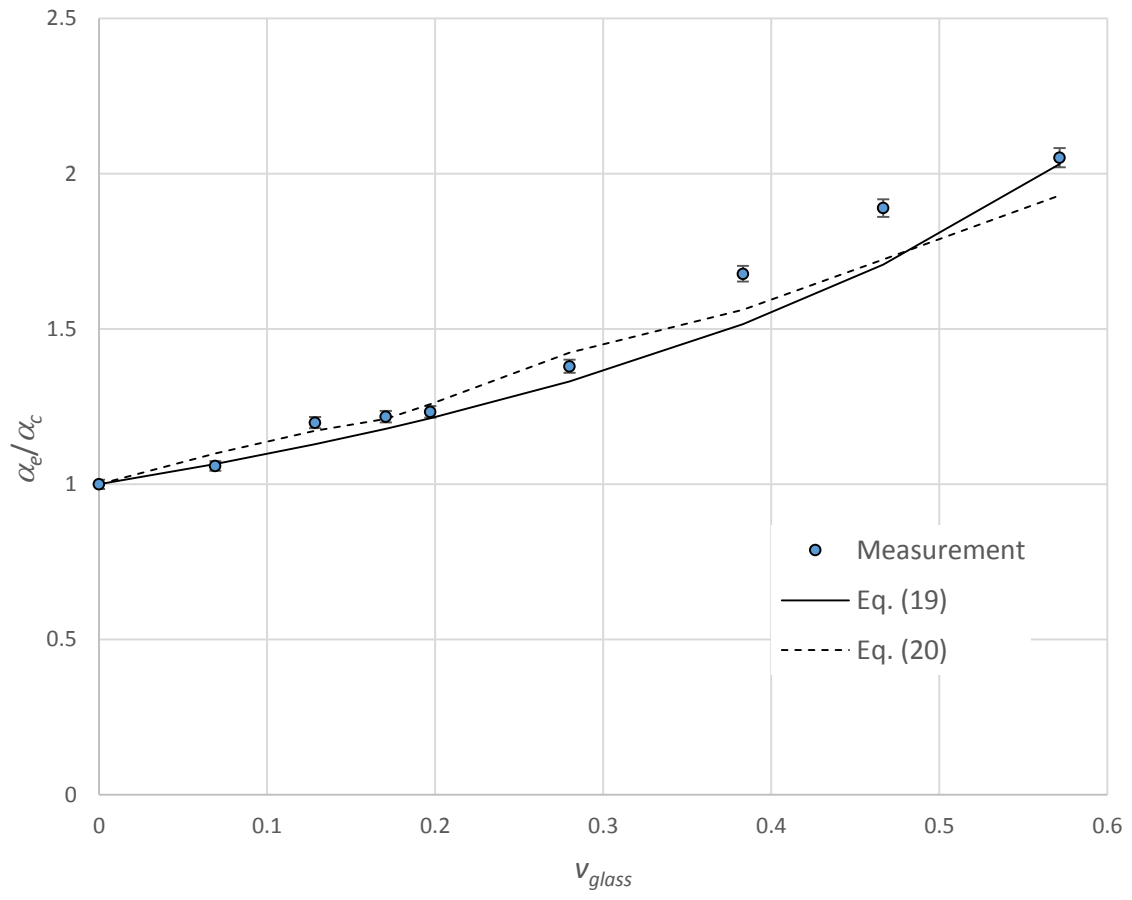


Figure 3

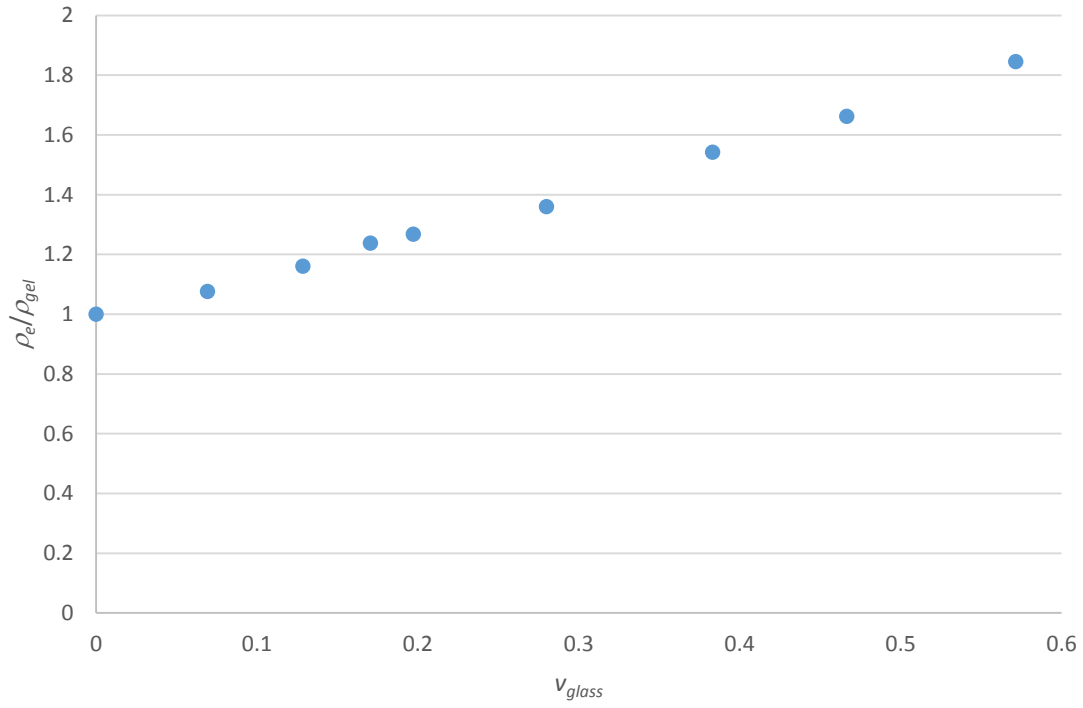


Figure 4

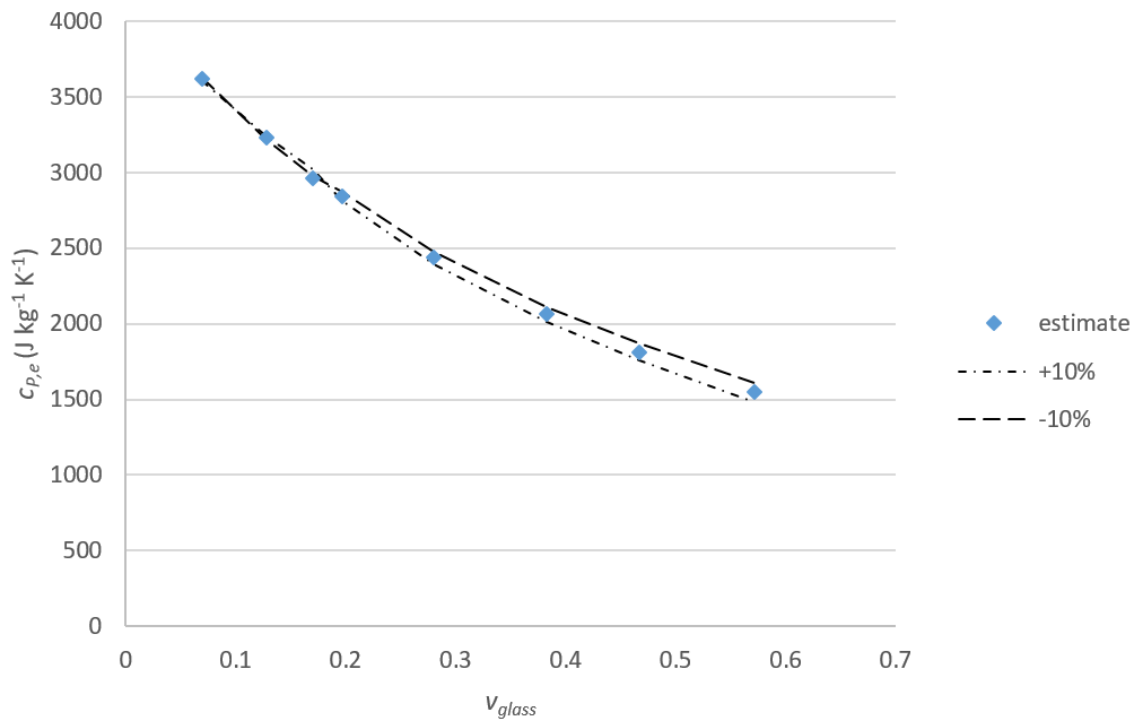


Figure 5

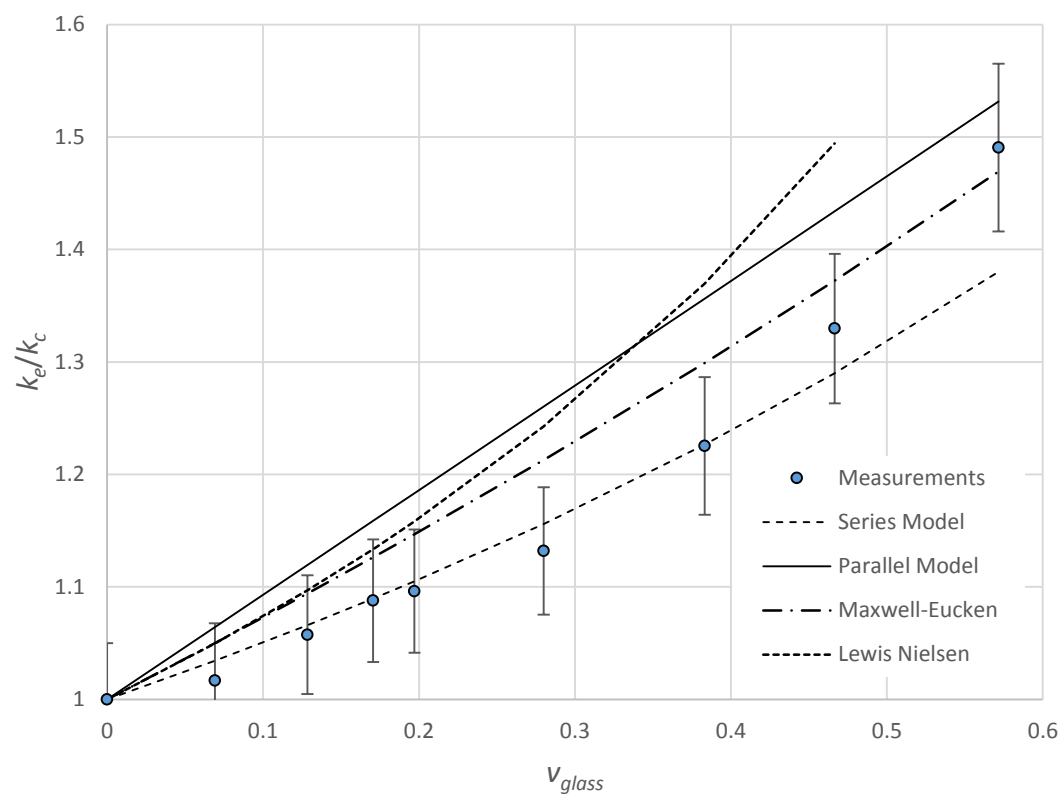


Figure 6